Semiconductor Electrodes. 44. Photoelectrochemistry at Polycrystalline p-Type WSe₂ Films

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Photoelectrochemical (PEC) cells (1) based on layered semiconductors (e.g., WSe,, MoSe,) are particularly attractive for solar energy conversion for a number of reasons. The excitation process involves d-d transitions and this may provide good stability for the photoelectrode. The band gaps of these materials are well-matched with the solar spectrum so that high efficiencies, in principle, can be realized. Finally, they are made of relatively abundant and inexpensive materials (e.g., W, Mo, S, Se). The pioneering work of Tributsch and The co-workers (2) on the use of these materials for regenerative photoelectrochemical cells has stimulated research in this area (3) and cells using small, selected single crystals have shown high efficiencies (e.g., 14% in red light (4) and 10.2% under solar irradiation).(5)

One of the major drawbacks of these systems, however, is that crystal imperfections and exposed edges on the surface perpendicular to the C axis (\bot, C) act as recombination centers for the photogenerated carriers or as sites for back-reaction of photogenerated products, thus decreasing efficiencies dramatically. (3b,h) As a result the observed efficiencies are strongly dependent on the nature of the surface of the individual $_2$ crystals and large area electrodes ($0.1 \ \ cm^2$) which show good efficiencies have not been produced. Our research group (6) and others have attempted to produce materials as electrodes by vapor deposition or sintering, but these generally showed no or very small photocurrents under conditions (e.g., with I^{I}/I_{2} electrolytes) where single crystals work well, presumably because of the abundance of surface imperfections and exposed edges. We report here the preparation of polycrystalline p-WSe electrodes which show better than 1% red ²light efficiencies in photovoltaic (regenerative) PEC cells.

Films of p-WSe, on quartz were prepared as follows. Tungsten selenide (WSe,) powder (Gallard Schlesinger 99.8%) was introduced into a quartz tube (15 cm long and 18 mm diameter) which had been cleaned successively

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with HF(48%), 1:1 HNO_2 :HCl by volume, HF(48%), then thoroughly rinsed with distilled water and dried. After evacuating for about 45 min to 1×10^{-5} torr, the tubes were sealed under vacuum and placed inside a Jelrus dental furnace at 850° C for a period of 67 h. The furnace was then turned off and the tube was allowed to come to ambient temperature. At this stage, part of the material had deposited on the walls of the tube, giving it a mirror-like appearance; the underside of the film had a black matte appearance. The tube was opened and pieces of the quartz with deposited WSe, were mounted as electrodes. This was done by contacting the front surface of the deposited material with conductive silver paint (Acme Industries, New Haven, Conn.) to a copper lead which was inserted into 6 mm bore glass tubing. The electrode was masked with 5-min epoxy cement except for the surface to be investigated; the areas investigated were typically 0.04 to 0.20 cm². In some cases it was possible to dislodge thin films of WSe, from the quartz tubes. These were mounted in the same fashion, except that back surface contacts were made. Electrochemical equipment and techniques were as previously reported. (7) Irradiation of the photoelectrodes was performed with an Oriel (Stamford, Conn.) 450 Watt Xe lamp with a water filter and a 590 nm gut-on filter. The focused power was ~ 88 mW/cm².

PEC Cells. The PEC behavior of these electrodes was investigated in cells containing redox couples; a number of Co(bpy)₃^{3†} (bpy = 2,2'-bipyriaine), (4,4'-dimethylbipyridinium or methyl viologen) (¥4. $Ru(NH_3)_{6}^{3+}$ and $Fe(L)^{3+}$ (where L is the macrocyclic ligand shown in Figure 1 prepared by condensation of diacetylpyridine and triethylenetetraamine as described by Curry and Busch.(8a) Table I is a compilation of the results obtained with these different ples. The highest output power was with the Fe(L) complex. The redox couples. obtained response of a polycrystalline electrochemical and 0.5 \underline{M} p-WSe₂ electrode in aqueous Fe(L) 0.3 <u>M</u> Na₂SO₄ under chopped illumination is

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	Redox Couple	Supp. Elect.	E ^{O'} vs. SCE(V)	v (a) oc (MV)	i (b) _@ mA/cm ²	E _{appl} . (V)
1)	Ru(NH ₃) ^{3+/2+}	0.67 <u>M</u> KC1	-0.18	137	3.8	0.25
					5	0.0
2)	Fe(CN) ₆ ^{3-/4-}	0.5 <u>M</u> Na ₂ SO ₄	+0.45	130	2.8	+0.4
3)	MV ^{2+/+}	0.6 <u>M</u> KC1	-0.69		0.5	-0.4
4)	Fe(EDTA) ^{1-/2-}	0.2 <u>M</u> Na ₂ SO ₄	-0.085		2	+0.2
5)	Co(bpy) ₃ ^{3+/2+}	0.15 <u>M</u> Na ₂ SO ₄	0.05	322	1.5	0.2
6)	Fe(Macro) ^{3+/2+}	0.5 <u>M</u> Na ₂ SO ₄	-0.10	480	i _{ph sc} =	6-7
				2		

Photoelectrochemical Behavior of p-WSe₂ Electrodes in the Presence of Table I: Different Redox Couples

V = open circuit photovoltage under 88 mW/cm² illumination. iph = photocurrent under 88 mW/cm² illumination and under ph = photocurrent control at the stipulated value (a) (b)

Short circuit photocurrent under 88 mW/cm⁻² illumination. (c)

shown in Figure 2. The potential for onset of photocurrent is about 0.40 V, which represents an underpotential of \sim 0.5V; E° for the Fe(L) couple is -0.10V at pH 4.0.(8b) There was little dark current over this potential range and the photocurrent density approached a limiting value of 10 mA/cm² at 0.0 V vs. SSCE. A two-electrode PEC cell containing 0.3 M Fe(L) and 5 mM Fe(L)²⁺ in aqueous 0.5 M Na $_2$ SO₄ with a large area Pt counter electrode showed short circuit photocurrent densities of 6 mA/cm² (geometric area) and open-circuit photovoltages of 480 The fill factor for the cell was only mV 🕳 0.28. This represents a solar energy conversion efficiency of about 1% in terms of the light incident on the cell wall (88 mW/cm²). A somewhat larger value would be obtained if corrections for reflectivity and solution absorbance were made.

Characterization of Polycrystalline <u>p-WSe_ Electrodes.</u> Characterization of the films involved the measurement of the flat band potential ($V_{\rm E}$), the apparent doping density in terms of the geometric area of the electrode, and film resistivity, as well as investigation of film appearance by scanning electron microscopy (SEM).

A Mott-Schottky plot $(1/C^2 vs. V)$ for a polycrystalline p-WSe electrode in aqueous 0.2 <u>M</u> Na₂SO₄ at a ²frequency of 1 kHz was linear (Fig. 3). The intercept yields a value of +0.78V vs. SCE for V_{fb}. This is in good

agreement with measurements in aqueous media (3g) on single crystal p-WSe, electrodes (V = 0.72V vs. SCE). From the stope of the plot, an apparent value of N, of 2.8 x 10^{-0} cm⁻³ can be calculated. The black matte surface of the film is quite rough, however, so that the actual acceptor density is probably of the order of $10^{18} \times 10^{10}$ cm⁻³. There was some frequency dispersion in the Mott-Schottky plots over frequencies of 100 Hz to 1 KHz, but the calculated values of V $_{\rm fb}$ and N were not affected significantly. The surface topography of these materials was investigated by SEM (Fig. 4). The surfaces appear to be composed of randomly oriented thin plates and in some cases scattered islands of small ($^{\circ}200$ $^{\mu}$ m) were observed. crystals The presence of these small crystalline islands were not necessary for the observation of a photoeffect, however, and some of the best results were obtained on samples that showed no such islands. The average film thickness, estimated from the weight of the deposit and the density of WSe (9.35 g/cm) was 35 μm . The resistance of the films across two front surface contacts 0.5 cm apart yielded values of 250 to 3000 ohms for different samples.

It is difficult for us to account for the success of these polycrystalline films as earlier photoelectrodes compared to The method of surface preparations. pretreatment of the substrate appears to be Similar photoeffects have been important.

found with over 60 electrodes prepared in four independent batches, SO that accidental special conditions of preparation or experimental artifacts appear unlikely. We have been informed that 99.999% WSe, will not transport under the conditions of²this work and that minor impurities, including residual water in the tube, may play an important role in the deposition of our material.(10) The manufacturer reports the following levels of impurities (in %): Al, 0.01; Cr, 0.01; Cu, 0.01; Fe, 0.05; Ni, Mn, Si, < 0.01; no halogens were detected. We have also detected Fe on the transported material by electron microprobe analysis. Perhaps the very large surface ([C) to edge ratio or fortuitous passivation of the edge planes in the preparation leads to the reasonable While the efficiency efficiencies observed. of these materials is still too small to be of practical interest, modifications of the method of preparation, the film pretreatment, solution conditions should lead to and significant improvements. For example the 0.3 \underline{M} Fe(L) solution is highly colored and absorbs substantial amounts of light. Several methods have also been suggested for passivating the edges of n-type layer compounds which have resulted in improved performance.(5,9) Finally, the tubular method of preparation may allow a simple direct fabrication of a PEC cell, where the photoelectrode is deposited on one side of the tube, which itself is the cell body, with a concentric internal counter electrode.

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Figure 1: Structure of macrocyclic ligand used.



Figure 3: Mott-Schottky plot (1/c² vs. E) for polycrystalline p-WSe₂ in aqueous 0.2 \underline{M} Na₂SO₄ in the dark at 1 KHz.





Figure 4: Scanning electron micrographs of polycrystalline p-WSe₂. (a) Low magnification (40X). Note small crystalline islands. (b) High magnification (4,000X).



Figure 2: Electrochemistry under chopped illumination for a polycrystalline p-WSe, electrode in aqueous .5 $\underline{\text{M}}$ Na SO containing .3 $\underline{\text{M}}$ Fe^{III}(Macro), .005 $\underline{\text{M}}$ Fe^{II}(Macro). Incident light $\sim 88 \text{ mW/cm}^2$.